

# Epitaxial wafer scale growth of tungsten dichalcogenides

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Transition metal dichalcogenides (TMDs) have been a focus of interest due to the direct band gap of the monolayers in the range of 1.6 – 2.0 eV and large exciton binding energies, which leads to interesting electronic and optical properties. One major challenge in harnessing the potential of these materials is the growth of high quality epitaxial mono- and few-layer films over large areas. The growth on C-plane sapphire is expected to lock the domain orientations at 0° and 60° due to the hexagonal symmetry, which can result in epitaxial TMD films with reduced high-angle grain boundaries.<sup>1</sup> Oriented growth for TMDs like MoS<sub>2</sub><sup>1</sup> and WSe<sub>2</sub><sup>2</sup> has been demonstrated by powder vapor transport on C-plane sapphire previously. However, we find that in a cold-wall metal organic chemical vapor deposition (MOCVD) process, even though this orientation relation is maintained for WSe<sub>2</sub>, this locking does not extend universally to WS<sub>2</sub>. The reason for this difference is crucial in not only obtaining oriented films but also understanding the basic interactions between the TMDs themselves and/or the precursors involved and the sapphire substrate.

In this work, WS<sub>2</sub> and WSe<sub>2</sub> mono- and few-layer films were deposited by MOCVD system on 2" C-plane sapphire wafers using tungsten hexacarbonyl (W(CO)<sub>6</sub>), hydrogen selenide (H<sub>2</sub>Se) and purified hydrogen sulfide (H<sub>2</sub>S). The growth was carried out for 1 h at 800- 900°C for WSe<sub>2</sub> and 850-1000°C for WS<sub>2</sub> to achieve fully coalesced films with domains on the order of 1 μm in size. The results show that there is a distinct difference in the growth of WSe<sub>2</sub> and WS<sub>2</sub> films. Though both WSe<sub>2</sub> and WS<sub>2</sub> have an epitaxial relation with the underlying sapphire substrate, the WSe<sub>2</sub> domains are predominantly oriented at 0° and 60°, but the WS<sub>2</sub> films show presence of domains at other angles as well, as shown in Figure 1. In case of WSe<sub>2</sub>, the orientation of 0° and 60° is maintained at all the growth temperatures investigated, but for WS<sub>2</sub>, the domains are oriented at 0° and 30° at lower temperatures between 750-850°C. Additional orientations emerge at 900-1000°C. Further details about the epitaxial relation, the interface interaction and the differences in the growth of WS<sub>2</sub> and WSe<sub>2</sub> will be presented.

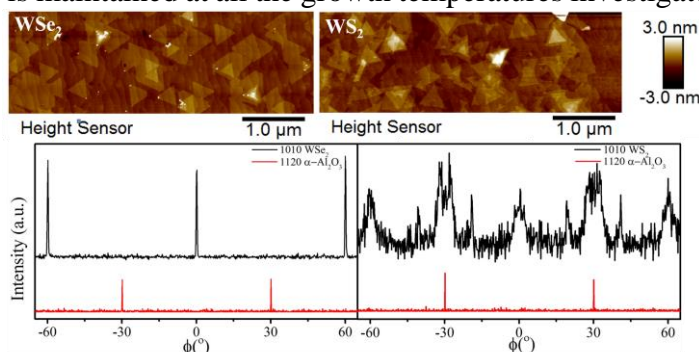


Figure 1: Atomic force microscopy images and the phi- scans highlighting the growth of WSe<sub>2</sub> oriented at 0° and 60° and the multiple orientations observed for WS<sub>2</sub>.

- [1] D. Dumcenco, D. Ovchinnikov, K. Marinov, P. Lazi, M. Gibertini, N. Marzari, O. L Sanchez, Y. C. Kung, D. Krasnozhan, M. W. Chen, S. Bertolazzi, P. Gillet, A. F. Morral, A. Radenovic, A. Kis, *ACS Nano*, 9(4), **2015**, 4611–4620.
- [2] L. Chen, B. Liu, M. Ge, Y. Ma, A. N. Abbas, C. Zhou\*, *ACS Nano*, 9 (8), **2015**, 8368–8375.

## Supplementary Page

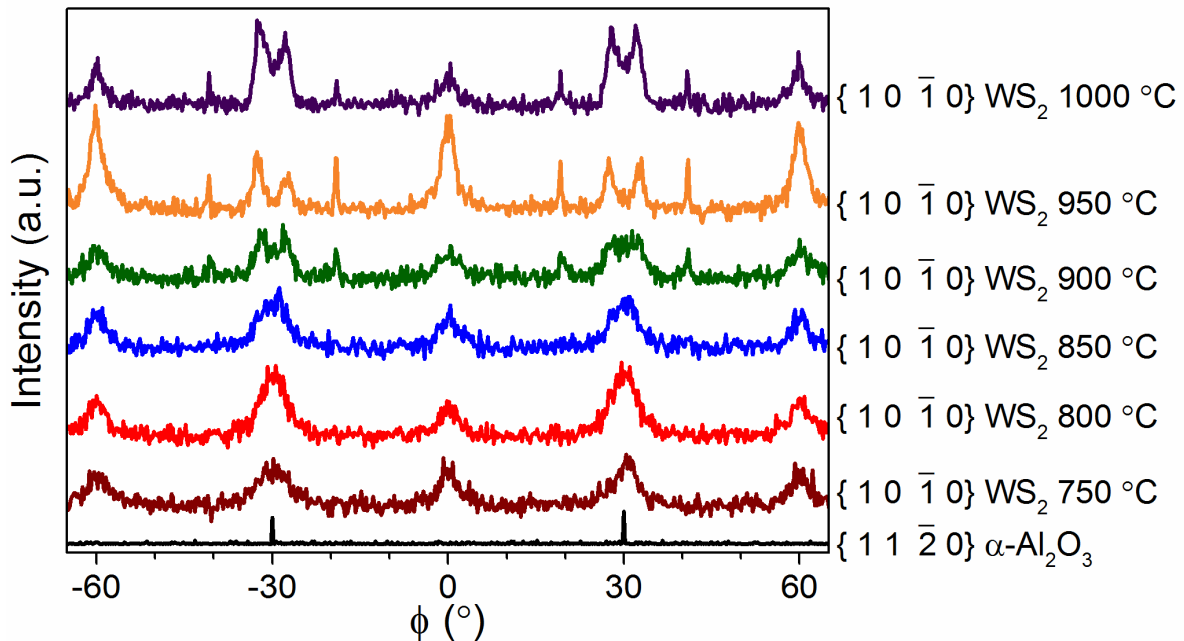


Figure 2 shows the evolution of the in-plane orientations of WS<sub>2</sub> as a function of the deposition temperature.

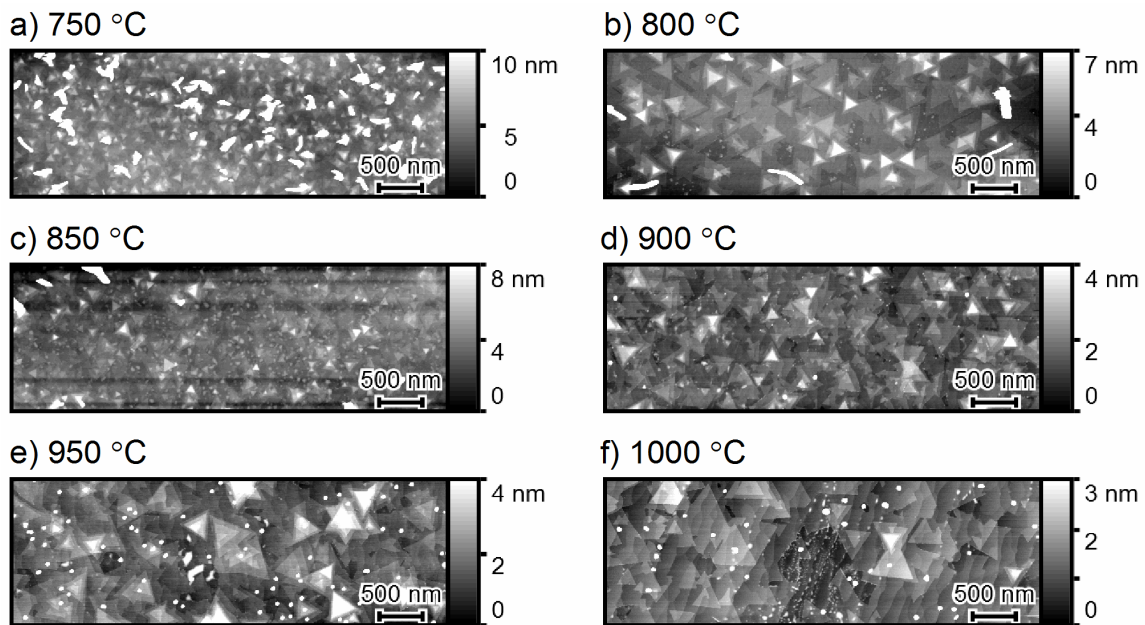


Figure 3 (a-f) shows the atomic force morphology of WS<sub>2</sub> as a function of the deposition temperature.

At lower temperatures, the films are oriented at 0 and 30° but at temperatures greater than 850°C, other orientations become prominent. The growth temperature plays a significant role in determining the orientation of growth.