

Bulk Growth

Room Bansal Atrium - Session BG-MoP

Bulk Growth Poster Session I

BG-MoP-1 MOCVD Development for Growth of Ga₂O₃ Over Large Areas, Muhammad Ali Johar, A. Feldman, G. Provost, K. Vasudevan, Structured Materials Industries, Inc; L. Lyle, Pennsylvania State University; L. Porter, Carnegie Mellon University, USA; A. Popp, Leibniz-Institut für Kristallzüchtung (IKZ); G. Tompa, Structured Materials Industries, Inc
Growth of high crystal quality gallium oxide (Ga₂O₃) at large production scale throughput is not available in the market due to (a) a ready high-volume supply of large wafers for homoepitaxy, (b) MOCVD tools explicitly configured for large area MOCVD growth of Ga₂O₃ homo- or heteroepitaxy. One of the primary factors is the unavailability of manufacturing scale substrates and growth tools.

Our company "Structured Materials Industries, Inc, (SMI)" has designed, sold, and fielded several Ga₂O₃ MOCVD growth tools as well as adapted several pre-existing (for different materials) tools to grow Ga₂O₃ and operated in-house tools for growth on single (cm² scale, 50mm, 100mm, 150mm, and 200mm) and multiple wafer tools (3×50mm, 19×50mm, and 38×50mm wafer diameters) or multiple 100mm) depositions. These tools have operated using a range of heating systems – induction, radiant filament, and lamps to operate over a range of growth temperatures (~400°C to ~1100°C) using a multiple of binary and alloy growth precursors (O₂, N₂O, H₂O, TEGa, TMGa, TMI_n, TMAI, TEAL, GaCl₃), process enhancers such as HCl, and dopants (SiH₄, TEOS, TMSn) and are compatible with a wide range of other elements.

The results of these works have led SMI to refine a series of reactors for research to production applications – spanning single to multiple wafer tools employing a multiple of growth parameters and heater types. In this presentation, we will review our series of reactor designs and how they are best applied – for example few through multiple small or large wafer epitaxy is best produced using advanced patented (US7573004 B1) high temperature oxide compatible designed filaments in rotating disc reactors supporting growth through 900°C whereas very high temperature growth – through 1100°C is best produced using induction heating when such high temperatures are warranted, whereas enclosed filament geometries are best for large single wafer processing. We note that filament heating is sufficient for most applications and can be used to produce uniform temperature heating. Reactor designs, supported by simulations of a variety of parameters, such as gas mixtures, flows, temperatures, pressures, carrier flows, flow rates of precursors, and the gap between showerhead and susceptor, for example, are reviewed along with areas of demonstrated results. Overall, production tool designs are based upon optimized parameters and they are reviewed.

BG-MoP-2 Quality Improvement of Sn-doped β-Ga₂O₃ Single Crystal by Optimizing Temperature Gradient Control in Growth Zone, Su-Min Choi, H. Jang, S. Seo, M. Chae, M. Park, Y. Jang, Department of Advanced Materials Engineering, Dong-Eui University, Republic of Korea; Y. Moon, Y. Sung, J. Kang, AXEL, Republic of Korea; Y. Shin, S. Bae, Korea Institute of Ceramic Engineering and Technology, Republic of Korea; W. Lee, Department of Advanced Materials Engineering, Dong-Eui University, Republic of Korea

As an ultra-wide bandgap semiconductor(UWBG), β-Ga₂O₃ is a very promising materials for various applications such as UV photodetectors, power rectifiers, gas sensors, MOSFETs, SBDs, and LEDs because β-Ga₂O₃ has a wide bandgap of 4.9eV and a high breakdown voltage of 8MV/cm. Furthermore, since β-Ga₂O₃ is capable of melt growth, it is possible to obtain a higher growth rate and a lower manufacturing cost than other WBG semiconductors such as SiC, GaN, and Diamond.[1,2]

In this study, the quality improvement of Sn-doped β-Ga₂O₃ crystal was systematically investigated with controlling temperature gradient in growth cell of edge-defined film-fed growth (EFG) method. Flat-shaped β-Ga₂O₃ crystal ribbons were typically grown by the EFG method. The principal surface and the growth direction were set to be (001) and [010], respectively. To prepare Sn-doped n-type bulk crystals, different amounts of Sn powder were added to Ga₂O₃ source powder. The temperature gradient in growth cell was controlled by changing structure of surrounding refractory materials and using the after-heater(A/H). The doping characteristics and crystal quality of various β-Ga₂O₃ crystal ribbons grown with changing the temperature gradient were systematically analyzed. A

phase analysis of β-Ga₂O₃ bulk crystals was performed by Raman analysis, and the quality of the crystals was analyzed by a high-resolution X-ray diffraction (HRXRD). Electrical properties and impurity concentration of unintentionally doped (UID) β-Ga₂O₃ and Sn-doped β-Ga₂O₃ crystals were analyzed using the Hall effect measurement system and the secondary ion mass spectrometry (SIMS), respectively.

High-quality Sn-doped (001) β-Ga₂O₃ crystal with a full width at half maximum (FWHM) of 38 arcsec and an etch pit density (EPD) of 3700/cm² was obtained with optimizing temperature gradient. Proper temperature gradient resulted in the reduction of SnO₂ volatilization during the growth process and the uniform distribution of bluish color throughout the entire ribbon. The carrier concentration of ≥10¹⁸/cm³ was obtained on Hall effect measurement.

Reference

- [1] S. Ohira et al., Thin Solid Films 516, 5763-5767 (2008)
- [2] Stephan Lany, APL Mater. 6, 046103 (2018)

BG-MoP-4 Various Crystal Planes and their Characteristics obtained from β-Ga₂O₃ Single Crystal Blocks Grown by the Multi-slit Structure of the EFG Method, Y. MOON, AXEL, Republic of Korea; HUIYEON JANG, Dongeui University, Republic of Korea; Y. SUNG, AXEL, Republic of Korea; S. CHOI, M. CHAE, S. SEO, M. PARK, Y. JANG, W. LEE, Dongeui University, Republic of Korea; Y. SHIN, S. BAE, Korea Institute of Ceramic Engineering and Technology, Republic of Korea; T. LEE, H. KIM, Korea Institute of Industrial Technology, Republic of Korea; J. KANG, AXEL, Republic of Korea

β-Ga₂O₃ 4.9eV [1]. It has a wide bandgap and a high breakdown voltage of 8MV/cm, so it is receiving a lot of attention for power device applications. In addition, it has the outstanding advantage of being able to grow a single crystal with a higher growth rate and lower manufacturing cost than other WBG semiconductor materials such as SiC, GaN, and diamond. [1-4] In this study, a β-Ga₂O₃ crystal block with a thickness of 10 mm was grown on an Ir die with a multi-slit structure through an edge-define film-fed growth (EFG) process. The growth direction and main surface of the β-Ga₂O₃ crystal were set to the [010] direction and the (100)/(001) plane, respectively, and the crystal growth rate was set to about 12 mm/h. The crystalline blocks were annealed in a nitrogen environment to reduce residual stress in the crystals before preparing crystal substrates with different orientations. A chemical mechanical polishing (CMP) process using Hastilite Ditron 3.0 slurry was chosen as the final processing step to create an epi-ready surface of the β-Ga₂O₃ crystal substrate. Raman spectra showed successful growth of β-Ga₂O₃ bulk crystals. Crystal quality was assessed by high-resolution X-ray diffraction (HRXRD). Etch pit density (EPD) and defect types were measured on crystal surfaces etched with H₃PO₄ solution.

Etch pit density (EPD) and defect type were measured on the liquid etched crystal surfaces. A typical Sn-doped Ga₂O₃ crystal block grown by the EFG method was bluish. The X-ray rocking curve values of the Ga₂O₃ single crystal ribbon depend on the Sn doping concentration and show slightly different values depending on the measurement position in the Ga₂O₃ crystal ribbon, indicating the spatial change of the Ga₂O₃ crystal quality. The crystal quality and surface properties of various β-Ga₂O₃ crystal substrates with (100), (001), (-201), and (hkl) orientations after CMP process were systematically investigated.

Reference:

- [1] JY Tsao 등, Adv. 전자. 임마. 4, 1600501 (2018)
- [2] M. Higashiwaki, GH Jessen, Appl. Physics Rhett. Man 112, 060401 (2018)
- [3] Kun Zhang 외, J. Alloys and Compounds, 881, 160665 (2021)
- [4] Shengnan Zhang 외, J. Semicond. 39, 083003 (2018)

Monday Evening, August 14, 2023

BG-MoP-5 Investigation of Defects in(100) and (001) β -Ga₂O₃Single Crystal GrownbyEFG Method, M. Choi, Korea Institute of Ceramic Engineering and Technology/Pusan National University, Republic of Korea; Yun-Ji Shin, Korea Institute of Ceramic Engineering and Technology, Republic of Korea; W. Jeong, T. Gu, A. Shin, S. Cho, Korea Institute of Ceramic Engineering and Technology/Pusan National University, Republic of Korea; Y. Moon, J. Kang, AXEL, Republic of Korea; W. Lee, Dong-Eui University, Republic of Korea; S. Jeong, Korea Institute of Ceramic Engineering and Technology, Republic of Korea; S. Harada, Nagoya University, Japan; K. Ishiji, Kyushu Synchrotron Light Research Center, Japan; H. Lee, Pusan National University, Republic of Korea; S. Bae, Korea Institute of Ceramic Engineering and Technology, Réunion

Gallium oxide (Ga₂O₃) has been highlighted as an emerging material for power semiconductor applications [1]. Recent achievements in bulk substrates and epitaxy pave a way for demonstrating several kV level power devices. However, various defects in β -Ga₂O₃ such as dislocations, stacking faults, and twin defects have been still issued, requiring further crystal quality with defect analysis. X-ray topography is one of powerful techniques to observe defect distribution in a non-destructive way [2]. The etch pit technique destructively visualize the defect distribution of the substrate as it chemically etches down the defect regions with an acid (or base). In this study, we compare the defect characteristics of (100) and (001) β -Ga₂O₃ single crystals grown by edge-defined film-fed growth (EFG).

In experiment, the etch pits of (100) and (001) β -Ga₂O₃ single crystals were formed by wet etching with 85 wt% phosphoric acid (H₃PO₄) solution at 140 °C for 120 min. The profile of etched surfaces was observed using atomic force microscope. In addition, (001)-oriented β -Ga₂O₃ single crystal was analyzed using an X-ray topography (Beamline 09, Synchrotron Radiation Facility SAGA-LS).

Figure 1 shows the AFM images of the etched surface for (100) and (001) surfaces. The etch pit of (100) plane are shallower and narrower compared to (001) plane as the (100) plane is parallel to the dislocation direction and has a cleavage property [3]. The density of the etch pits in the (100) and (001) planes were counted to be $\sim 3.8 \times 10^5 \text{ cm}^{-2}$ and $\sim 6.0 \times 10^4 \text{ cm}^{-2}$, respectively. Figure 2 shows the XRT image of (001) β -Ga₂O₃ with a diffraction vector, $g=605$. It was found that the dominant features were dot, line along [010] direction, and curved line contrasts. The line contrasts are mostly associated with b-axis screw dislocations with Burgers vectors (b) parallel to [010], and the dot contrasts are related to edge dislocations with $\langle 001 \rangle$ burgers vectors [4]. In addition, many wandering dislocations are observed on the (001) surface, which show curved contrasts. This defect observation might assist to improve the crystal quality of Ga₂O₃ substrates as we perform further in-depth characterization in the future.

References

- [1] J. Y. Tsao, et al., *Adv. Electron. Mater.* 4, 1600501 (2018)
- [2] M. Higashiwaki, et al., *Appl. Phys. Lett.* 100, 013504 (2012).
- [3] Kun Zhang, et al., *J. Alloys and Compounds*, 881,160665 (2021)
- [4] Shengnan Zhang, et al., *J. Semicond.* 39, 083003 (2018)

Tuesday Afternoon, August 15, 2023

Epitaxial Growth

Room Davis Hall 101 - Session EG+BG-TuA

Bulk/Epitaxy II

Moderator: Sriram Krishnamoorthy, University of California Santa Barbara

1:45pm **EG+BG-TuA-1 Suitable Orientation for Homoepitaxial Growth of Gallium Oxide**, *Kohei Sasaki, A. Kuramata*, Novel Crystal Technology, Inc., Japan
INVITED

The surface orientation is an important condition in homoepitaxial growth. β -Ga₂O₃ has an unusual crystal structure, named β -Gallia, so we cannot use the knowledge of the usual crystal structures, such as diamond, zinc blende or wurtzite, when the selecting the surface orientation. Here, we investigated suitable orientations for homoepitaxial growth of gallium oxide by growing films on gallium oxide substrates with various orientations.

The β -Gallia structure is monoclinic, and its low index planes are (100), (010), and (001). We made gallium oxide substrate with surface orientations from the (100) to (010) plane or from the (100) to (001) plane and investigated the crystal quality, surface roughness, and growth rate of the films grown by molecular beam epitaxy. We sliced the growth in ten degrees steps from the (100) plane rather than adjusting the specific orientation. Growth temperature was fixed at 700 degrees Celsius. Ozone gas was used as the oxygen source.

Of the planes between the (100) and (010) plane, only the (100) plane showed a peculiarly low growth rate. On the other hand, there were no unusual features on the planes except the (100) plane; the growth rate was about 700 nm/h, and surface roughness (RMS) was about 1-2 nm.

On the other hand, the planes between the (100) and (001) plane showed severe surface roughness especially around the (101) plane and (-201) plane. The surface roughness on the (101) plane was due to crystal defects in which (-201) crystal grew on the (101) plane, whereas on the (-201) plane it was due to (-201) twin defects. We obtained very smooth surfaces with an RMS of 1 nm or less by using the (001), (-102), (401), (-401) planes.

It is known that the surface orientation of gallium oxide homoepitaxial growth depends on the growth method. The surfaces of films grown by MBE and metalorganic chemical vapor deposition (MOCVD) show similar morphologies. Thus, the knowledge gained in this research may be applicable to MOCVD.

2:15pm **EG+BG-TuA-3 Pushing the Al composition limit up to 99% in MOCVD β -(Al_xGa_{1-x})₂O₃ films using TMGa as Ga precursor**, *A F M Anhar Uddin Bhuiyan, L. Meng, H. Huang, J. Hwang, H. Zhao*, The Ohio State University

Recent research progresses have highlighted the promising potential of the MOCVD growth method in developing β -(Al_xGa_{1-x})₂O₃ alloys along different crystal orientations with high Al composition and controllable n-type doping. The coexistence of β and γ phases in (010) β -AlGaO films with Al>27% indicates challenges for incorporating higher Al compositions. Using alternative crystal planes of β -Ga₂O₃ substrates, such as (100) and (-201), has yielded single phase β -AlGaO films, with over 50% of Al incorporation. These prior efforts on MOCVD growth of β -AlGaO alloys using TEGa as the Ga precursor limit the film growth rate to below ~0.7 μ m/hr.

In this study, we employed TMGa as the Ga precursor, which not only elevates the growth rates of β -AlGaO films up to 2 μ m/hr, but also enhances the Al compositions up to a record high value of ~99%. The systematic investigation of MOCVD growth of β -AlGaO films and β -AlGaO/Ga₂O₃ superlattices on different crystal planes revealed a strong impact of substrate orientation on the solubility limit of β -AlGaO grown at relatively high growth rates. The crystalline structure, strain, morphology, stoichiometry, and bandgap of β -AlGaO films are investigated as a function of the Al composition and crystal orientations. β -AlGaO films with Al compositions up to 99%, 29%, 16% are achieved on (100), (010) and (-201) β -Ga₂O₃ substrates, respectively, as determined by XRD, XPS and STEM EDX. Beyond 29% of Al incorporation, the (010) β -AlGaO films exhibit β to γ phase transformation. Owing to its highly anisotropic characteristics, (-201) β -AlGaO films show local segregation of Al. Fully strained coherent β -AlGaO films are obtained for thicknesses of 350 nm (010, Al=15%), 120 nm (100, Al=16%) and 205 nm (-201, Al=13%). The crystalline structure of 20 nm thick β -(Al_{0.99}Ga_{0.01})₂O₃ film was accessed by atomic resolution STEM imaging, showing sharp interface and alloy homogeneity. The electron nano-diffraction pattern and quantitative STEM-EDX elemental mapping confirm the β -phase growth with Al composition of 99%, which agrees well with XRD and XPS measurement results. A record high bandgap energy of 7.26 eV is achieved from β -(Al_{0.99}Ga_{0.01})₂O₃ film using XPS, revealing great

promises of developing β -AlGaO/Ga₂O₃ interfaces with high band offsets. The findings of this study offer valuable insights on the MOCVD epitaxy and properties of high Al composition β -AlGaO films and β -AlGaO/Ga₂O₃ heterostructures for device applications.

Acknowledgment: AFOSR (FA9550-18-1-0479) and NSF (Grant No. 2231026, and 2019753).

2:30pm **EG+BG-TuA-4 Fast Growth and Characterization of Undoped β -Ga₂O₃ on 2-Inch Substrates Using a Horizontal Hot-Wall MOVPE System**, *Kazutada Ikenaga*, Tokyo University of Agriculture and Technology / TAIYO NIPPON SANSO CORPORATION, Japan; *J. Yoshinaga, P. Guanxi*, TAIYO NIPPON SANSO CORPORATION, Japan; *H. Tozato, T. Okuyama, K. Goto, Y. Kumagai*, Tokyo University of Agriculture and Technology, Japan
Metalorganic vapor phase epitaxy (MOVPE) is one of the attractive methods for the epitaxial growth of β -Ga₂O₃. However, it requires control of the hazardous reactions between organometallics and oxygen (O₂) while suppressing the incorporation of carbon (C) and hydrogen (H) impurities derived from the organometallics. Our research group has clarified the key conditions that enable the growth of high-purity β -Ga₂O₃ layers with suppressed C and H incorporation by thermodynamic analysis and in situ mass spectrometry of gaseous species in the reactor [1-3]. In this work, we report on the uniform and fast growth of β -Ga₂O₃ on 2-inch substrates.

A horizontal low-pressure hot-wall MOVPE system (TAIYO NIPPON SANSO CORPORATION, FR2000-OX) with a facedown holder capable of placing 2-inch diameter substrates was used. One 2-inch diameter c-plane sapphire wafer or three 10 mm × 15 mm sized β -Ga₂O₃(010) substrates were set for each growth. Epitaxial layers were grown in a temperature range of 900–1050 °C using trimethylgallium (TMGa) and O₂ as precursors, and Ar as a carrier gas, respectively. Under a constant O₂ supply, TMGa was supplied in the range of 111 – 546 μ mol/min (corresponding to the input VI/III ratio from 1609 to 327).

The growth rate was found to be constant regardless of the growth temperature. At a growth temperature of 1000°C, the growth rate increased linearly up to about 15 μ m/h with increasing TMGa supply rate, while the C impurity concentration increased. Since an increase in H and C impurity concentrations was observed with decreasing growth temperature, it is likely that the increase in these impurities is due to the increase in TMGa-derived hydrocarbons and their insufficient combustion. It was also found that there is no difference in growth rate between heteroepitaxial growth and homoepitaxial growth under the same conditions. In this presentation, the uniformity of the grown layer is also reported.

This work was supported by Ministry of Internal Affairs and Communications (MIC) research and development (JPMI00316).

[1] K. Goto et al., Jpn. J. Appl. Phys. **60**, 045505 (2021).

[2] K. Ikenaga et al., J. Cryst. Growth **582**, 126520 (2022).

[3] K. Ikenaga et al., Jpn. J. Appl. Phys., in press.

2:45pm **EG+BG-TuA-5 MBE Growth and Properties of Ultra-wide Bandgap Oxide Layers Spanning 5.0 - 9.0 eV Energy Gaps**, *Debdeep Jena*, Cornell University
INVITED

3:15pm **EG+BG-TuA-7 Structural Defect Formation and Propagation in Fe-doped Czochralski-grown b-Ga₂O₃ Boules**, *Luke Lyle*, Pennsylvania State University - Applied Research Lab; *R. Lavelle*, Penn State University - Applied Research Lab; *D. Erdely*, Pennsylvania State University - Applied Research Lab; *W. Everson*, Penn State University - Applied Research Lab; *A. Balog, N. Alem*, Pennsylvania State University; *D. Snyder*, Pennsylvania State University - Applied Research Lab

Over the last decade, β -Ga₂O₃ has garnered increased attention due to its ultrawide bandgap of 4.7-4.9 eV, controllable range of shallow, n-type dopants (Sn, Si, Ge), and easily scalable and economic melt-growth processes. Popular melt-growth processes that have been developed for β -Ga₂O₃ include the edge-defined film fed method, vertical bridgeman, and the Czochralski process. Although different types of structural defects in these melt-grown crystals have been identified, how they form and propagate throughout the growth process remains elusive. Specifically, it has been found that the density of structural defects can vary across wafers in the same boule and even across a single wafer.

We etch and analyze double side, chemi-mechanically polished 2" diameter wafers and 1" diameter wafers taken from 'cores' from 2" diameter boules at the tip, center, and tail of Fe-doped (010) Czochralski-grown boules. The etch pits were formed using an optimized H₂PO₄ etch process and are

Tuesday Afternoon, August 15, 2023

mapped using automated optical microscopy, statistical analysis software and scanning electron microscopy and are organized by defect type and density across each wafer analyzed in the boule. Wafers from the length of the boule were used to assess seeding and growth initiation related defect structures and long-range propagation and results from adjacent wafers at various locations were studied to understand short range defect formation and propagation. Trends regarding the presence of dislocations/nanopipes and their formation throughout the boule are discussed along with differentiation between process- and growth-related defects. Particular attention in this talk is paid to the formation and propagation of so-called "nanopipe" defects, as they are poised to act as killer-defects for high-voltage devices

Bulk Growth

Room Bansal Atrium - Session BG-TuP

Bulk Growth Poster Session II

BG-TuP-5 β -Ga₂O₃ Single Crystal Growth by EFG Method using Die with Multi-Slit Structure, *Yeon-Geun Seong, Y. Moon, Axel, Republic of Korea; H. Jang, S. Choi, C. Min-Ji, S. Seo, M. Park, Y. Jang, W. Lee, Dongeui University, Republic of Korea; J. Kang, Axel, Republic of Korea*

β -Ga₂O₃ is attracting attention as a next-generation power semiconductor. β -Ga₂O₃ has a high bandgap of 4.9eV and a high breakdown voltage of 8MV/cm. In addition, β -Ga₂O₃ grown by the EFG (Edge Defined Film-Fed Growth) method is superior to other power semiconductor materials such as SiC and GaN due to its fast growth rate and low manufacturing cost. However, since the β -Ga₂O₃ crystal grown by the EFG method grows in a ribbon morphology, the number of wafers that can be extracted from one ingot is small. [1-4]

In this study, the thickness of the ingot was increased through a die with multi-slit structure. Crystal growth from multi-slit structure is divided into 'diameter direction', which determines the size of the wafer, and 'thickness direction', which determines the extraction numbers of wafer. As a result of β -Ga₂O₃ growth experiments using Multi-Slit Die, we found that thick crystal growth is difficult if the growth rate in the diameter direction is too fast, and polycrystals are easily to occur if the growth rate in the thickness direction is too fast. Therefore, in order to overcome these problems, the two-dimensional temperature distribution and the temperature gradient in the vertical direction were adjusted to secure reproducibility to stably grow thick crystal with high crystallinity.

As a result of the experiment, various process conditions, such as the type and structure of insulation, three-dimensional temperature gradient, and pulling speed, had a more sensitive effect on the growth of thickness direction in multi-slit die compared with single-slit die. By adjusting the thermal balance of upper and lower parts of crucible and the temperature gradient of die in diameter direction and thickness direction, the growth of thick β -Ga₂O₃ single crystals was successfully achieved. This result can contribute to lower the manufacturing cost of Ga₂O₃ crystals as a substrate for power semiconductor fabrication.

Reference

- [1] J. Y. Tsao et al, Adv. Electron. Mater. 4, 1600501 (2018)
- [2] M. Higashiwaki, G.H. Jessen, Appl. Phys. Lett. 112, 060401 (2018)
- [3] Kun Zhang et al, J. Alloys and Compounds, 881,160665 (2021)
- [4] Shengnan Zhang et al, J. Semicond. 39, 083003 (2018)

Epitaxial Growth

Room Davis Hall 101 - Session EG+BG+MD-WeM

Epitaxial III

Moderators: Hari Nair, Cornell University, Uttam Singiseti, University of Buffalo, SUNY

9:15am **EG+BG+MD-WeM-4 Growth of α -(Al_xGa_{1-x})₂O₃ by Suboxide Molecular-Beam Epitaxy**, Jacob Steele, K. Azizie, N. Pieczulewski, J. McCandless, D. Muller, H. Xing, D. Jena, Cornell University; T. Onuma, Kogakuin University, Japan; D. Schlom, Cornell University (USA) and Leibniz-Institut für Kristallzüchtung (Germany)

Ga₂O₃ has attracted significant interest due to its ultra-wide bandgap, high electron mobility, and large breakdown field. These properties exceed the current benchmarks set by materials such as SiC and GaN, making Ga₂O₃ optimal for next-generation power devices. Still, it has been proposed that the properties of Ga₂O₃ can be extended further by alloying with Al to form (Al_xGa_{1-x})₂O₃ which can raise the bandgap to 8.6 eV. This goal presents a challenge for the most researched phase, β , as β -Ga₂O₃ thermodynamically prefers a monoclinic structure and α -Al₂O₃ is stable in the corundum structure. This structural mismatch limits the compositional range and the range of attainable bandgaps. In contrast, α -Ga₂O₃ occupies the corundum structure and has been shown to alloy over the full compositional range, enabling bandgaps from 5.3 - 8.6 eV. One method of growing α -(Al_xGa_{1-x})₂O₃ is molecular-beam epitaxy (MBE). MBE is a powerful and highly controllable growth technique for α -(Al_xGa_{1-x})₂O₃ thin films with drawbacks being slow growth rates of a few hundred nm/h and narrow adsorption-controlled growth windows. One method to improve the growth rate is the technique of suboxide MBE, which allows growth of β -Ga₂O₃ thin films at rates exceeding 1 μ m/h with large adsorption-controlled growth regimes.

We show that suboxide MBE can be used for the epitaxial growth of high quality α -(Al_xGa_{1-x})₂O₃ thin films on A plane sapphire substrates over the full range of x at greater than 1 μ m/h. For our study, gallium suboxide, Ga₂O, and elemental Al are the MBE sources. The oxidant is 80% distilled ozone which is held at constant pressure (5 \times 10⁻⁶ Torr) while the Ga₂O and Al fluxes are varied to control composition. We measure the composition of our films with XRD and confirm that we cover the full range of 0 < x < 1 with vacuum ultraviolet transmittance measurements showing that the bandgaps of our films shift from α -Ga₂O₃ to α -Al₂O₃. We show that the film composition can be controlled directly by the relative ratios of the Ga₂O and Al fluxes. Our films have high structural quality as revealed by the full width at half maximum (FWHM) of rocking curves of the α -(Al_xGa_{1-x})₂O₃ films ranging from 11 - 15 arcseconds; these FWHMs are identical to the underlying sapphire substrates. The surfaces of the films are also smooth with RMS roughnesses measured by atomic force microscopy ranging from 0.3 - 1.1 nm on α -(Al_xGa_{1-x})₂O₃ films with thicknesses in the 17.8 - 47.8 nm range. We also show our progress with growing α -(Al_xGa_{1-x})₂O₃ films over 100 nm thick and with doping using a SiO₂ source.

9:30am **EG+BG+MD-WeM-5 Structural, Electrical, and Thermal Characterization of CIS-MOCVD β -Ga₂O₃ Epitaxial Buffer Layers**, Hannah Masten, Naval Research Laboratory; G. Alvarez, Cornell University; C. Halverson, Washington State University; M. Liao, J. Lundh, Naval Research Laboratory; F. Alema, A. Osinsky, Agnition Technology; A. Jacobs, Naval Research Laboratory; M. Weber, Washington State University; Z. Tian, Cornell University; K. Hobart, M. Tadjer, Naval Research Laboratory

Epitaxial growth of β -Ga₂O₃ using metalorganic chemical vapor deposition (MOCVD) has seen great advancements demonstrating high-quality films with low point defect concentrations and high mobility with low doping concentrations [1]. Here, we investigate the impact of buffer layer thickness for these MOCVD epitaxial films on electrical characteristics, thermal conductivity, and defect concentrations.

MOCVD films were grown on Novel Crystal Technology's Fe-doped (010) β -Ga₂O₃ substrates using Agnition Technology's Agilis close-injection showerhead MOCVD (CIS-MOCVD). The unintentionally doped (UID) buffer layer thickness was varied on the 3 samples: A-300, B-500, and C-1000 nm. The UID layers were followed by a 10 nm thick n⁺ (\sim 10¹⁹ cm⁻³) Ga₂O₃ layer for improved channel conductivity. A 100 nm highly n⁺ layer was selectively regrown following ref. [2]. Ohmic contacts were formed in the regrown areas with an annealed 20/200 nm Ti/Au metal stack (470 °C, 1 min., N₂). Mesa isolation was formed with an etch of \sim 170 nm. Transmission line measurements (TLM) showed sample C had the lowest specific contact resistance of 2.25 \times 10⁻⁶ Ω -cm² and sample A had the highest of 1.99 \times 10⁻⁴ Ω -cm². Room temperature Hall effect measurement showed similar

mobility for B and C of 115-116 cm²/V-s, while sample A showed a much lower mobility of 71 cm²/V-s. Samples B and C, both showed high open-gated source-drain current (I_b) (>0.05 A/mm at V_{DS}= 5 V) and low isolation (mesa-mesa) current (I_{iso}) of < 0.1 μ A/mm at V_{DS}= 10 V. Sample A (300 nm thick buffer layer), showed 10X lower open-gated I_b and a high I_{iso} of \sim 3 mA/mm at V_{DS}= 10 V. Higher I_{iso} for samples with thin buffer layers, such as sample A, have been frequently attributed to a peak in Si concentration at the epilayer/substrate interface observed in secondary-ion mass spectroscopy [1]. Here, we offer further insight on this effect via frequency-domain thermoreflectance (FDTR) and positron annihilation spectroscopy (PAS). Preliminary FDTR data showed decreasing thermal conductivity for thicker epilayers. PAS data fitted with a 3-layer model consistently showed higher density of Ga-related vacancies in the epilayers compared to each substrate. More detailed measurements, including XRD and device-level FDTR, will be performed. This preliminary data suggested that MOCVD Ga₂O₃ was affected by both unintentional impurities and point defects in addition to the known issue of interfacial Si accumulation. [1] A. Waseem, et al., *Physica Status Solidi (A)*, p. 2200616, 2022. [2] Z. Xia, et al., *IEEE EDL*, 39(4), 568-571, 2018.

9:45am **EG+BG+MD-WeM-6 Electrical and Optical Properties of Melt-Grown Mn Doped β -Ga₂O₃**, Benjamin Dutton, C. Rempel, J. Jesenovec, Washington State University; J. Varley, L. Voss, Lawrence Livermore National Laboratory; M. McCluskey, J. McCloy, Washington State University

Several acceptor dopants have been explored in β -Ga₂O₃ to produce semi-insulating substrates and epitaxial films. Fe and Mg make up the majority of research thus far, however, other transition metals provide potential alternatives for optimized performance. β -Ga₂O₃ bulk single crystals were grown by the Czochralski and vertical gradient freeze methods with a nominal dopant concentration of 0.25 at.% Mn. Ultraviolet-visible-near infrared spectroscopy and photoluminescence revealed polarization and orientation dependent optical absorptions and a unique orange luminescence. All samples were electrically insulating, indicative of acceptor doping on the order of 10⁹ - 10¹¹ ohm-cm at room temperature. Actual dopant concentrations of the intentionally doped transition metal and background impurities were determined via glow discharge mass spectrometry, indicating the macro-scale segregation behavior. Laser- ablation inductively-coupled plasma mass spectrometry along with photoluminescence mapping revealed micro-scale segregation of impurity ions. Density functional theory calculations were carried out to elucidate likely site-occupancy and the acceptor level of Mn in the band gap.

10:00am **EG+BG+MD-WeM-7 Mg and Zn Counter doping of Homoepitaxial β -Ga₂O₃ Grown by Molecular Beam Epitaxy**, Stephen Schaefer, K. Egbo, S. Harvey, A. Zakutayev, B. Tellekamp, National Renewable Energy Laboratory

Gallium oxide has attracted attention as a candidate material for high-power diodes and transistors owing to its wide bandgap and high breakdown voltage. Homoepitaxial β -Ga₂O₃ has been successfully grown by plasma-assisted molecular beam epitaxy, however it is well-documented that unintentional Si donors at the epitaxial interface lead to the formation of an undesirable parasitic conducting channel. Mg and Zn are deep acceptor levels in β -Ga₂O₃ and Mg counterdoping by MBE has been shown to compensate unintentional donor impurities. However counterdoping with other elements such as Zn remains sparsely investigated.

We report on Mg and Zn counterdoping in homoepitaxial β -Ga₂O₃ grown by MBE on (010) Fe-doped (semi-insulating) and (001) Sn-doped (n-type) wafers. A valved cracker source is used for Mg while Zn is evaporated from a conventional effusion cell. Mg- and Zn-doped stacks are measured by secondary ion mass spectroscopy to calibrate the cell temperatures and valve positions to the dopant incorporation. A typical Ga₂O₃ growth temperature is 600 °C and growth rates are 0.47 - 0.70 Å/s. β -Ga₂O₃ samples composed of a \sim 2 nm Mg- or Zn-doped layer and a 300 nm unintentionally doped layer are grown with dopant fluxes ranging from 3.8 \times 10⁻⁹ to 2.0 \times 10⁻⁸ torr. Counterdoped samples grown on (001) Sn-doped and (010) Fe-doped wafers are processed into vertical and lateral Schottky devices, respectively. In both devices the Ohmic contact is formed by stable 5 nm Ti / 100 nm Au annealed under N₂ at 550 °C while the Schottky contact is formed by 30 nm Ni / 100 nm Au. The Schottky devices are characterized by capacitance-voltage (C-V) measurements at 20 kHz.

We find that the C-V characteristics of the vertical Schottky devices grown on (001) Sn-doped Ga₂O₃ show a reduction in residual capacitance and corresponding increase in depletion width at high reverse bias voltage for the Mg-counterdoped sample compared to an undoped control sample grown under identical conditions. Additionally, the I-V characteristic of the Mg doped device exhibits lower reverse leakage current. These findings are

Wednesday Morning, August 16, 2023

mirrored in lateral Schottky devices grown on (010) Fe-doped Ga₂O₃ where counterdoping with 1.0×10⁻⁸ torr Zn flux results in approximately ~2× reduction of capacitance and effective carrier concentration while counterdoping with the same Mg flux results in ~5× reduction. The C-V results suggest that Mg and Zn effectively compensate unintentional donors in Ga₂O₃. Experiments including an annealing study of Mg and Zn diffusion in β-Ga₂O₃ are expected to yield insight to the controllability of counterdoping in Ga₂O₃.

10:15am **EG+BG+MD-WeM-8 Optimizing Si Implantation and Annealing in β-Ga₂O₃**, *Katie Gann*, N. Pieczulewski, Cornell University; T. Asef, Air Force Research Laboratory; C. Gorsak, Cornell University; K. Heinselmann, national renewable Energy Laboratory; K. Smith, J. McCandless, Cornell University; B. Noesges, Air Force Research Lab; G. Xing, D. Jena, H. Nair, D. Muller, M. Thompson, Cornell University

Optimizing the thermal anneal of Si implanted β-Ga₂O₃ is critical for low resistance contacts and selective area doping in advanced device structures. We report the impact of annealing time, temperature, and ambient on the activation of ion-implanted Si in β-Ga₂O₃ at concentrations from 5×10¹⁸ to 1×10²⁰ cm⁻³, and in β-(Al_xGa_{1-x})₂O₃ (x≤15%) at 5×10¹⁹ cm⁻³. Nearly full activation (>90%) and high mobilities (>70 cm²/V-s) are achieved in β-Ga₂O₃ with contact resistances below 0.16 Ω-mm. In β-(Al_xGa_{1-x})₂O₃, initial results are promising with moderate activation (50%) and high mobility (60 cm²/V-s).

UID β-Ga₂O₃ films were grown by plasma assisted MBE on Fe-doped (010) β-Ga₂O₃ substrates; comparable β-(Al_xGa_{1-x})₂O₃ films were grown by MOCVD. Si was implanted at multiple energies to yield 65 or 100 nm box profiles with concentrations of 5×10¹⁸, 5×10¹⁹, or 1×10²⁰ cm⁻³. To understand damage accumulation, low and high temperature implants were also studied. Anneals were performed in a UHV-compatible quartz furnace at 1 bar with well-controlled gas ambients.

To maintain β-Ga₂O₃ stability, P_{O₂} must be greater than 10⁻³ bar (based on annealing in vacuum or forming gas). For 5×10¹⁹ cm⁻³ Si, full activation is achieved for P_{O₂}<10⁻⁴ bar while 5×10¹⁸ cm⁻³ tolerates ~10⁻² bar. Water vapor is critical even at 1 ppm; at 25 ppm active carriers are reduced by 10x. Optimal results were obtained with H₂O below 10 ppb. Based on recovery with subsequent “dry” anneals, we propose an OH-mediated defect compensating Si dopants.

Lattice recovery (mobility) occurs for T > 900 °C, with carriers and mobility increasing with temperature to 1050 °C. However, SIMS shows substantial Si diffusion above 1000 °C with 950 °C the optimal anneal temperature. Activation at 950 °C is maximized between 5 and 20 minutes with shorter times exhibiting slightly lower mobilities while longer times result in carrier deactivation; this “over-annealing” behavior occurs at all temperatures and becomes more significant at high concentrations. Room temperature implants to 1×10²⁰ cm⁻³ are shown to fully activate under these optimal conditions.

To understand lattice damage recovery, implants at varying temperatures were characterized by XRD, Rutherford Backscattering Channeling (RBS/C), and STEM. XRD showed no second phases under any conditions. RBS/C and STEM showed only partial amorphization with remnant aligned β-Ga₂O₃. We propose a model to explain the efficient activation based on 3D lattice recovery in the absence of full amorphization.

Author Index

Bold page numbers indicate presenter

— A —

Alem, N.: EG+BG-TuA-7, 3
Alema, F.: EG+BG+MD-WeM-5, 6
Alvarez, G.: EG+BG+MD-WeM-5, 6
Asel, T.: EG+BG+MD-WeM-8, 7
Azizie, K.: EG+BG+MD-WeM-4, 6

— B —

Bae, S.: BG-MoP-2, 1; BG-MoP-5, 2
BAE, S.: BG-MoP-4, 1
Balog, A.: EG+BG-TuA-7, 3
Bhuiyan, A.: EG+BG-TuA-3, 3

— C —

Chae, M.: BG-MoP-2, 1
CHAE, M.: BG-MoP-4, 1
Cho, S.: BG-MoP-5, 2
Choi, M.: BG-MoP-5, 2
Choi, S.: BG-MoP-2, 1; BG-TuP-5, 5
CHOI, S.: BG-MoP-4, 1

— D —

Dutton, B.: EG+BG+MD-WeM-6, 6

— E —

Egbo, K.: EG+BG+MD-WeM-7, 6
Erdely, D.: EG+BG-TuA-7, 3
Everson, W.: EG+BG-TuA-7, 3

— F —

Feldman, A.: BG-MoP-1, 1

— G —

Gann, K.: EG+BG+MD-WeM-8, 7
Gorsak, C.: EG+BG+MD-WeM-8, 7
Goto, K.: EG+BG-TuA-4, 3
Gu, T.: BG-MoP-5, 2
Guanxi, P.: EG+BG-TuA-4, 3

— H —

Halverson, C.: EG+BG+MD-WeM-5, 6
Harada, S.: BG-MoP-5, 2
Harvey, S.: EG+BG+MD-WeM-7, 6
Heinselman, K.: EG+BG+MD-WeM-8, 7
Hobart, K.: EG+BG+MD-WeM-5, 6
Huang, H.: EG+BG-TuA-3, 3
Hwang, J.: EG+BG-TuA-3, 3

— I —

Ikenaga, K.: EG+BG-TuA-4, 3

Ishiji, K.: BG-MoP-5, 2

— J —

Jacobs, A.: EG+BG+MD-WeM-5, 6
Jang, H.: BG-MoP-2, 1; BG-TuP-5, 5

JANG, H.: BG-MoP-4, 1

Jang, Y.: BG-MoP-2, 1; BG-TuP-5, 5

JANG, Y.: BG-MoP-4, 1

Jena, D.: EG+BG+MD-WeM-4, 6; EG+BG+MD-WeM-8, 7; EG+BG-TuA-5, 3

Jeong, S.: BG-MoP-5, 2

Jeong, W.: BG-MoP-5, 2

Jesenovec, J.: EG+BG+MD-WeM-6, 6

Johar, M.: BG-MoP-1, 1

— K —

Kang, J.: BG-MoP-2, 1; BG-MoP-5, 2; BG-TuP-5, 5

KANG, J.: BG-MoP-4, 1

KIM, H.: BG-MoP-4, 1

Kumagai, Y.: EG+BG-TuA-4, 3

Kuramata, A.: EG+BG-TuA-1, 3

— L —

Lavelle, R.: EG+BG-TuA-7, 3

Lee, H.: BG-MoP-5, 2

LEE, T.: BG-MoP-4, 1

Lee, W.: BG-MoP-2, 1; BG-MoP-5, 2; BG-TuP-5, 5

LEE, W.: BG-MoP-4, 1

Liao, M.: EG+BG+MD-WeM-5, 6

Lundh, J.: EG+BG+MD-WeM-5, 6

Lyle, L.: BG-MoP-1, 1; EG+BG-TuA-7, 3

— M —

Masten, H.: EG+BG+MD-WeM-5, 6

McCandless, J.: EG+BG+MD-WeM-4, 6; EG+BG+MD-WeM-8, 7

McCloy, J.: EG+BG+MD-WeM-6, 6

McCluskey, M.: EG+BG+MD-WeM-6, 6

Meng, L.: EG+BG-TuA-3, 3

Min-Ji, C.: BG-TuP-5, 5

Moon, Y.: BG-MoP-2, 1; BG-MoP-5, 2; BG-TuP-5, 5

MOON, Y.: BG-MoP-4, 1

Muller, D.: EG+BG+MD-WeM-4, 6;

EG+BG+MD-WeM-8, 7

— N —

Nair, H.: EG+BG+MD-WeM-8, 7

Noesges, B.: EG+BG+MD-WeM-8, 7

— O —

Okuyama, T.: EG+BG-TuA-4, 3

Onuma, T.: EG+BG+MD-WeM-4, 6

Osinsky, A.: EG+BG+MD-WeM-5, 6

— P —

Park, M.: BG-MoP-2, 1; BG-TuP-5, 5

PARK, M.: BG-MoP-4, 1

Pieczulewski, N.: EG+BG+MD-WeM-4, 6; EG+BG+MD-WeM-8, 7

Popp, A.: BG-MoP-1, 1

Porter, L.: BG-MoP-1, 1

Provost, G.: BG-MoP-1, 1

— R —

Remple, C.: EG+BG+MD-WeM-6, 6

— S —

Sasaki, K.: EG+BG-TuA-1, 3

Schaefer, S.: EG+BG+MD-WeM-7, 6

Schlom, D.: EG+BG+MD-WeM-4, 6

Seo, S.: BG-MoP-2, 1; BG-TuP-5, 5

SEO, S.: BG-MoP-4, 1

Seong, Y.: BG-TuP-5, 5

Shin, A.: BG-MoP-5, 2

Shin, Y.: BG-MoP-2, 1; BG-MoP-5, 2

SHIN, Y.: BG-MoP-4, 1

Smith, K.: EG+BG+MD-WeM-8, 7

Snyder, D.: EG+BG-TuA-7, 3

Steele, J.: EG+BG+MD-WeM-4, 6

Sung, Y.: BG-MoP-2, 1

SUNG, Y.: BG-MoP-4, 1

— T —

Tadger, M.: EG+BG+MD-WeM-5, 6

Tellekamp, B.: EG+BG+MD-WeM-7, 6

Thompson, M.: EG+BG+MD-WeM-8, 7

Tian, Z.: EG+BG+MD-WeM-5, 6

Tompa, G.: BG-MoP-1, 1

Tozato, H.: EG+BG-TuA-4, 3

— V —

Varley, J.: EG+BG+MD-WeM-6, 6

Vasudevan, K.: BG-MoP-1, 1

Voss, L.: EG+BG+MD-WeM-6, 6

— W —

Weber, M.: EG+BG+MD-WeM-5, 6

— X —

Xing, G.: EG+BG+MD-WeM-8, 7

Xing, H.: EG+BG+MD-WeM-4, 6

— Y —

Yoshinaga, J.: EG+BG-TuA-4, 3

— Z —

Zakutayev, A.: EG+BG+MD-WeM-7, 6

Zhao, H.: EG+BG-TuA-3, 3