Wednesday Afternoon, July 1, 2020

Nanostructure Synthesis and Fabrication Room Auditorium - Session NS1-WeA

2D Nanomaterials by ALD II

Moderators: Jeffrey W. Elam, Argonne National Laboratory, Hyungjun Kim, Yonsei University, Korea

1:00pm NS1-WeA-1 Atomistic Simulation of ALD of 2D Transition-Metal Dichalcogenides, *Mahdi Shirazi, E Kessels, A Bol,* Eindhoven University of Technology, Netherlands INVITED

Extensive research has been done during the last decade to unravel the remarkable electronic properties [1], [2] of two dimensional transitionmetal dichalcogenides (2D-TMDs) in the monolayer regime. In spite of their astounding electrical properties, these material systems are not ready yet for replacing Si based materials for future nanometer-sized electronic devices. One key challenge is the integration of these materials in bottomup processes at low temperature (usually < 500 °C) into the semiconductor manufacturing flow. Horizontal growth at wafer scale with a large grain size (typically $1x1 \ \mu m^2$) is required for nano-electronic devices [3]. The cyclic process of atomic layer deposition (ALD) [4] with tight control over the chemical reactions shows promise as such a bottom-up process. The chemical reactions of ALD are self-limiting and are designed to proceed only at the surface. In this contribution, we have employed density functional theory (DFT) to provide fundamental insight into the reaction mechanisms of the MoS₂ growth. We have studied the deposition of MoS₂ that is initiated by the exposure of metal precursor Mo(NMe₂)₂(NtBu)₂ $(C_{12}H_{30}N_4Mo)$ to the SiO₂ surface and then followed by exposure of H_2S/H_2 as co-reagent in the second pulse [5]. In this so-called hetero-deposition, the involved chemical reactions during ALD lead to the formation of a buffer layer at the surface of SiO₂ [6]. After formation of this buffer layer, ALD enters into the steady-growth regime (also called homo-deposition). In the steady growth regime, vertically or horizontally aligned MoS₂ structures grow in a layer-by-layer fashion. The calculated reaction energies and activation energies indicate that the reaction kinetics in the heterodeposition are slower than reaction kinetics in homo-deposition. Artificial intelligence is used to generate an efficient interatomic potential using the calculated energies and forces of configurations obtained by DFT. The generated interatomic potential will be used for larger scale simulations to provide further fundamental insight into the deposition of MoS₂ by ALD.

[1] M. Chhowalla, H. S. Shin, G. Eda, L.-J. Li, K. P. Loh, H. Zhang, *Nat. Chem.*, **2013**, *5*, 263.

[2] K. F. Mak, C. Lee, J. Hone, J. Shan, T. F. Heinz, *Phys. Rev. Lett.*, **2010**, DOI:10.1103/PhysRevLett.105.136805.

[3] H. Kwon, et. al., *npj 2D Mater. Appl.*, **2019**, DOI:10.1038/s41699-019-0091-9.

[4] S. M. George, Chem. Rev., 2010, DOI:10.1021/cr900056b.

[5] M. Shirazi, W. M. M. Kessels, A. A. Bol, Phys. Chem. Chem. Phys., 2018, DOI:10.1039/C8CP00210J.

[6] M. Shirazi, W. M. M. Kessels, A. A. Bol, *APL Mater.*, **2018**, DOI:10.1063/1.5056213.

1:45pm NS1-WeA-4 ALD of MoSe₂ using New Precursors, *Raul Zazpe*, University of Pardubice, Czech Republic; *R Krumpolec*, Brno University of Technology, Czech Republic; *J Charvot*, *L Hromadko*, *H Shopa*, *M Motola*, *M Krbal*, *F Bures*, *J Macak*, University of Pardubice, Czech Republic

2D semiconductor transition metal dichalcogenides (TMDs) have attracted considerable attention due to their layered structure, suitable band gap for visible light absorption, high carrier mobility, electrochemically active unsaturated edges and relatively good stability against photocorrosion [1]. Recently, 2D MoSe₂ has been gaining considerable interest due to its higher electrical conductivity as compared to MoS₂, its wider inter-layer distance (~0.65 nm), narrow bandgap (1.33-1.72 eV), high resistance to photocorrosion, high surface area layer, electrochemically active unsaturated Seedges and close to zero Gibbs free energy edges for hydrogen adsorption. These properties are promising for different applications of MoSe₂ including hydrogen evolution [2], photocatalysis [3] and Li-ion batteries [4]. However, their low light absorption efficiency, recombination issues of the photogenerated electron-hole pairs and slow charge transfer of the intrinsic semiconducting 2H-phase are a handicap. An efficient strategy to surpass those intrinsic limitations are hybrid nanostructures using conducting supporting materials. In this regard, anodic ${\rm TiO}_2$ nanotubes (TNTs) are excellent photoactive supporting material providing a

high surface area, unique directionality for the charge separation, and highly effective charge collection. [5] Accordingly, we present anodic TiO₂ nanotubes homogenously decorated with MoSe₂ nanosheets by atomic layer deposition (ALD). In parallel, we address the current scarcity of convenient ALD Se precursors by the synthesis a set of new selenium precursors - alkysily! (R₃Si)₂Se and alkytin (R₃Sn)₂Se, and cyclic silylselenides compounds. Those Se precursors were extensicely characterized and their reliability as ALD Se precursors explored [6,7]. Several compounds exhibited promising results to be convenient ALD Se precursor as will be presented in the presentation. The synthesis of the MoSe₂ nanosheets and their composites with TiO₂ NTs, their physical and electrochemical characterization, and encouraging results in electrochemical characterization, hydrogen evolution reaction (HER) and photocatalysis will be presented and discussed.

2:00pm NS1-WeA-5 Low Temperature Creation of Layered-MoS₂ Thin Films on Large Area High Aspect Ratio Substrates, Anil Mane, D Choudhury, S Letourneau, J Elam, Argonne National Laboratory

Thin layers of two dimensional (2D) materials mainly transition metal dichalcogenides (TMDs) and more specifically ultra-thin layered- MOS_2 semiconductor possess exceptional properties such as electrical, optical, magnetic, mechanical and chemical properties. This allows the exploration of internal quantum degrees of freedom of electrons and their potential for use in semiconductor microelectronics, optoelectronic, energy, and sensor and detector applications. These exciting results are being achieved mostly by using exfoliation of flecks from bulk MOS_2 crystal. However, the biggest challenge in realizing TMDs full potential has been the lack of scalable material synthesis methods for such films with high uniformity, conformality and interfacing with other materials such as oxides, metals and its process compatibility.

Among the various thin film deposition methods, atomic layer deposition (ALD) offers the best combination of precisely controlled layer-by-layer thin film growth at low temperature with very high conformality on complex substrates. Here we will present the growth of layered-MoS₂ thin films. To grow high quality layered-MoS2 thin films, we have developed an ALDbased two step processing approach [1]: firstly the growth of well controlled ultra-thin layer of Mo metal using ALD followed by the sulfurization of the ALD Mo layer at various temperatures. This two-steps processing results in high quality layered-MoS₂ thin films on large substrates. For the Mo ALD process we used molybdenum hexafluoride (MoF₆) and Si₂H₆ precursors. We used in-situ QCM measurements to study interfacial and nucleation effects in the formation of continuous ultra-thin metal layer of Mo. The composition of both the ALD Mo and the layered MoS₂ layers was determined by X-ray photoelectron spectroscopy (XPS). Further, cross-sectional transmission electron microscopy (TEM) was performed to confirm the formation of layered MoS2 on high aspect ratio trenches and Raman analysis to verify the signature of E12g blue shift and A_{1g} red shift in the MoS₂ structure. In this presentation we will discuss the details of the two-step thin film growth process for creating layered MoS₂ layers via ALD Mo and subsequent sulfurization as well as the properties of the MoS₂ films.

[1] Anil Mane, Devika Choudhury, Steven Letourneau, Jeffrey Elam, (US patent application submitted 2018)

2:15pm NS1-WeA-6 Gas Sensing Characteristics of Mo_xW_{1-x}S₂ Synthesized by Atomic Layer Deposition, *Inkyu Sohn*, *Y Kim*, *M Lee*, *J Park*, *H Kim*, Yonsei University, Republic of Korea

Two-dimensional Transition metal dichalcogenide (2D TMDC) have been attracted great attention as gas sensing materials with high sensitivity in room temperature. [1] Because of this characteristic, 2D TMDC gas sensor could overcome the oxide-based semiconductor which need heating for gas sensing. Therefore, various 2D TMDC gas sensor studies have been ongoing. Recently, it has been shown that the gas sensor property could be improved through the changing the composition in WS_{2x}Se_{2-2x} alloy. [2]

Here we report a synthesis method of $Mo_xW_{1,x}S_2$ alloys for gas sensor by atomic layer deposition. Layer controlled 2D MOS_2 and WS_2 were synthesized with $Mo(CO)_6$, $W(CO)_6$ and H_2S as precursors and reactant. For the first time, we systematically modulate the composition of $Mo_xW_{1,x}S_2$ alloys by changing the configuration of low-temperature ALD super-cycles. AFM and Raman spectroscopy results of $Mo_xW_{1,x}S_2$ alloys demonstrate that the thickness of alloy is accurately controlled by ALD. Also XPS results confirmed that composition of alloy is precisely controlled by ALD supercycles. Furthermore, gas sensors fabricated by $Mo_xW_{1,x}S_2$ alloys were evaluated for NO_2 gas. It showed response time and recovery characteristics of $Mo_xW_{1,x}S_2$ alloy gas sensor is dramatically enhanced.

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Interfaces 10, 34163-34171 (2018)

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